

Large Area Graphene Field Effect Devices with Sputtered Nitride and Oxide Top-Dielectrics

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Realizing high-speed (high f_t) graphene transistors requires the high capacitance of a thin top-gated dielectric structure. The deposition of a top-gate dielectric can significantly reduce mobility due an increased density of trapped charges [1] and damage to the graphene during the deposition process. However, high- κ dielectrics such as ionic liquids have been observed to enhance mobility [2], explained by dielectric screening of charged impurities. The effect of a dielectric on mobility remains not fully understood [3,4]. Moreover, a graphene field effect device is doped by adsorbates according to the chemical nature of the dielectric surface in contact with the graphene [5]. We report an experimental study on the field effect mobility and doping of large area graphene field effect devices encapsulated by two top-layer dielectrics: SiO₂ ($\epsilon_r \sim 3.9$) and Si₃N₄ ($\epsilon_r \sim 3.9$) grown by magnetron sputter deposition.

Graphene was grown on Cu foils in a tube furnace with a methane/hydrogen gas mixture at a growth temperature of 900°C. The graphene was transferred to insulating substrates using a poly(methyl methacrylate) (PMMA) handle and sacrificial etch of the Cu in ammonium persulfate (0.1M). Further details of the growth and transfer process are described elsewhere [6,7]. Heavily doped Si (n-doped, 8-20 m Ω -cm) with 300 nm of dry, chlorinated thermal SiO₂, was chosen as the substrate for back-gated electrical measurements. Standard photolithography, electron beam evaporation and lift-off techniques were used to define Ohmic contacts (10nm Ti/100nm Au). Subsequent photolithography and oxygen plasma treatment was used to isolate graphene sheets in a 4-point van der Pauw geometry (Fig. 1).

Raman spectroscopy at a 514.5nm pump wavelength was used to characterize the graphene before and after deposition of a top-side dielectric. The sheet resistance R_S was measured by the van der Pauw technique as a function of back gate voltage V_{GS} . All electrical measurements were performed in a variable environment probe station following a 400K, 10⁻⁵Torr anneal over a minimum of 12 hours. Sheet resistance measurements were taken before and after deposition of a top-side dielectric. Magnetron sputtering with a Si target and RF plasma source was used to deposit 10nm of SiO₂ (20sccm of Ar and O₂ gas at 5mbar, 20nm/min growth rate) or 10nm of Si₃N₄ (25sccm of Ar and N₂ gas at 5mbar, 3.3nm/min growth rate).

Raman spectroscopy reveals an increase in D-peak ($\sim 1350\text{cm}^{-1}$) intensity following deposition of either dielectric, as well as an increase in background fluorescence (Fig. 2). The origin of the fluorescence is possibly due to plasma induced damage of graphene during sputtering [8]. Sheet resistance R_S versus back gate voltage V_{GS} measurements (Fig. 3) reveal that both SiO₂ and Si₃N₄ top-dielectric layers significantly reduce the doping of the field effect transistors, presumably due to the reduced presence of water [5]. The carrier mobility, $\mu = [CR_S(V_{GS} - V_{NP})]^{-1}$ where $C = 11.8\text{nF/cm}^2$ is the back-gate capacitance and V_{NP} is the neutrality point back-gate voltage, was compared at an electron density $n = 10^{12}/\text{cm}^2$ for a series of devices before and after deposition of top-gate dielectrics (Fig. 4). Carrier mobilities as high as $\sim 2000\text{cm}^2/\text{Vs}$ were achieved with a top-dielectric. Our results are comparable to that of microscopic exfoliated graphene samples encapsulated with nitride by chemical vapour deposition [9]. In several devices, a minimal change in mobility was observed before and after Si₃N₄ deposition. Future work to incorporate top-gated large-area graphene devices in high-frequency compatible structures is underway.

References

- [1] M.C. Lemme et al., Solid State Electron. **52** (2008) 514-518.
- [2] F. Chen, J. Xia, and N. Tao, Nano Lett. **9**, (2009) 1621-1625.
- [3] L.A. Ponomarenko et al., Phys. Rev. Lett. **102**, (2009) 206603.
- [4] C.R. Dean et al., Nature Nanotech. **5**, (2010) 722-726.
- [5] P.L. Levesque et al., Nano Lett. **11**, (2011) 132-137.
- [6] A. Guermoune et al., under review.
- [7] X. Li et al., Science **324** (2009) 1312-1314.
- [8] T. Gokus et al., ACS Nano **3** (2009) 3963-3968.
- [9] W. Zhu, D. Neumayer, V Perebeinos and P. Avouris, Nano Lett. **10** (2010) 3572-3576.

Figures

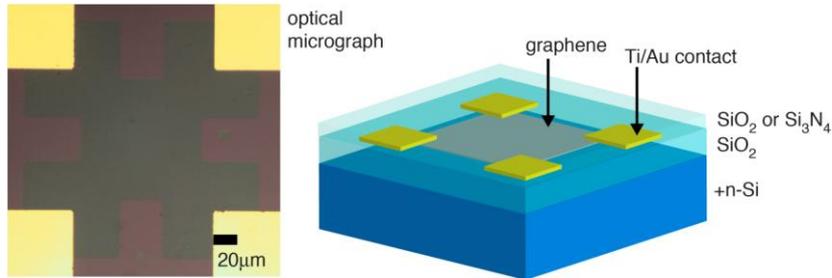


Fig. 1 : Graphene is transferred, etched and contacted into a van der Pauw sample on a SiO₂/Si substrate for back gating. A dielectric top layer of 10nm SiO₂ or 10nm Si₃N₄ is sputtered for encapsulation.

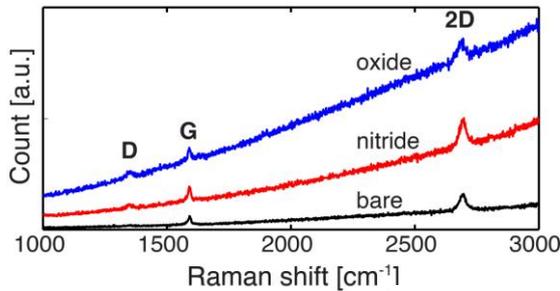


Fig. 2 : Raman spectra ($\lambda=514.5\text{nm}$) of large area graphene as transferred to SiO₂/Si. Following deposition of a Si₃N₄ dielectric top layer, an increase in D-peak ($\sim 1350\text{cm}^{-1}$) scattering and background fluorescence is observed. A more intense D-peak and greater background fluorescence is observed with SiO₂ deposited as a dielectric top layer.

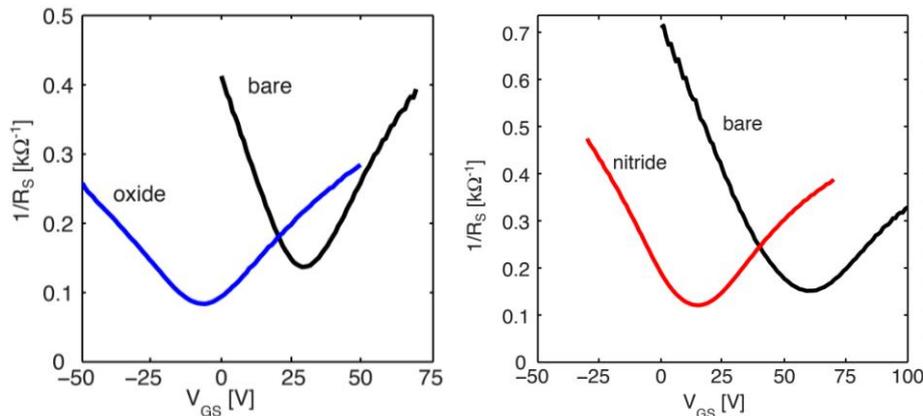


Fig. 3 : Sheet resistance R_S measured by van der Pauw technique versus back-gate voltage V_{GS} . A device before and after SiO₂ top-dielectric deposition (left) shows a reduction in doping and a reduction in mobility. A device before and after Si₃N₄ deposition (right) shows a reduction in doping and a negligible change in mobility.

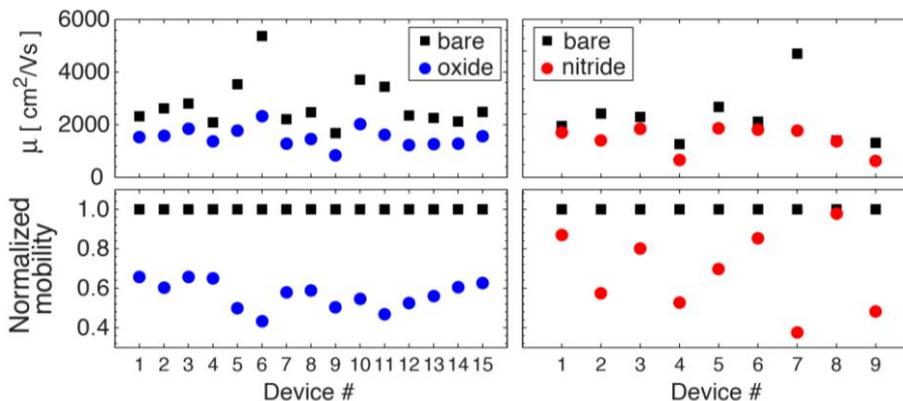


Fig. 4 : Carrier mobility μ at electron density $n = 10^{12}/\text{cm}^2$ inferred from R_S versus V_{GS} . The mobility is compared before and after dielectric deposition. The mobility normalized to that of the bare device before deposition is also compared. In several cases, a minimal change in mobility is observed.